

Synthesis and characterization of iron oxide nanoparticles reinforced polyester/nanocomposites

G. Naveen Kumar^{*1}, Y. V. Mohana Reddy², K. Hemachandra Reddy³

¹Department of Mechanical Engineering, S.K.University College of Engineering & Technology, SV Puram, Kadiri Road, Anantapuramu-515655, Andhra Pradesh, India

²Department of Mechanical Engineering, G Pulla Reddy Engineering College, Nandyal Road, Kurnool - 518 007, Andhra Pradesh, India.

³Department of Mechanical Engineering, Jawaharlal Nehru Technological University Anantapur, Anantapur-515002, Andhra Pradesh, India

*E-mail address: naveeng_phd@ymail.com, yvmr2006@gmail.com, konireddy@gmail.com

Keywords: Nanocomposites; f-Fe₂O₃ nanoparticles; Mechanical properties

ABSTRACT. We report on synthesis of two highly disperse nanoparticles viz. Fe₂O₃ & f-Fe₂O₃ using chemical reduction method. Reaction was initiated to mix solution A (i.e. FeCl₃ 6H₂O) into solution B (i.e. FeCl₂4H₂O) together under the presence of ammonium to develop nanoparticles. Mechanical properties of above mentioned nanoparticles filled with polyester polymer nanocomposites were fabricated to assess the possibility of using this filler as a new material. Methacryloxypropyl was used as a functionalization agent to prepare f-Fe₂O₃ nanoparticles. Mechanical properties of f-Fe₂O₃ nanocomposites were improved with the help of functionalization when compared with Fe₂O₃ nanocomposites. A functionalization of nanoparticles favours the composite fabrication with a lower curing temperature as compared to the as-synthesised nanoparticles filled polyester nanocomposites. Thermogravimetric analysis showed an increased thermo-stability of f-Fe₂O₃ nanoparticles filled polyester nanocomposites as compared to the Fe₂O₃ nanoparticles filled counterparts. Mechanical and thermal properties were increased due to the homogeneous particle dispersion and chemical bonding between nanoparticles and polyester matrix. The nanoparticles become magnetically harder after incorporation into the polyester resin matrix.

1. INTRODUCTION

Organic materials with both conducting and ferromagnetic properties have received tremendous attention due to their potential applications in batteries, electrochemical display devices, molecular electronics, nonlinear optics, sensors, electrical- magnetic shields and microwave-absorbants. Besides blending method by mixing physically inorganic ferromagnetic powder with conducting polymer powder, several approaches such as electrochemical and in-situ chemical polymerization have been reported to prepare conducting polymer film with ferromagnetic properties and nanocomposites with coreshell structure (Chen, et al., 2005; Podlaha and Landolt, 1997). Nanomaterials dramatically different from their bulk or atomic counterparts have attracted much interest due to their unique physicochemical properties for a wide range of potential device applications such as UV lasers, solar cells, high-sensitivity chemical gas or volatile organic compound sensors, and DNA sequence sensors (Castro, et al., 2000; Wetzel, et al., 2003; Guo, et al., 2007). Nanostructural materials such as nanoparticles (NPs) or nanofibers have been used as fillers in both the polymeric nanocomposites to improve the mechanical, electric, electronic and optical properties, and the metallic nanocomposite to control the electrodeposition. Polymer nanocomposites reinforced with inorganic nanoparticles have attracted much interest due to their lightness, homogeneity, cost-effective processability, and tunable physical properties (Kohli, et al., 2005; Judeinstein and Sanchez, 1996; Zhang, et al., 2005; Gao and Mader, 2002). Polyester resin, as a structural polymer, was chosen as a polymer matrix in current study due to the fact that the cured resins are thermosetting with a network structure possessing high resistance to the moisture and chemicals, and good mechanical properties. Thus the resultant composites have the potential

applications in fabrication and building materials such as electrodeposition tanks, automotive parts and marine vessels which require superior mechanical properties and/or high resistance to harsh environments such as strong acid or base. Furthermore, the functional groups of the polymer surrounding the nanoparticles enable these nanocomposites as good candidates for various applications such as site-specific molecule targeting in biomedical areas (Shenhar, et al., 2005; Guo, et al., 2006; Guo, et al., 2007a; Guo, et al., 2007b). The existing challenges in the composite fabrication are to obtain uniform filler dispersion and to introduce strong chemical bonding between the nanoparticles and the polymer matrix, which are necessary to provide a high tensile strength due to local stress within the nanocomposite. The interfacial interactions between fillers and polymer matrix play a crucial role in determining the quality and properties of the nanocomposites (Rancourt and Taylor, 2008; Sawada et al., 2008; Chiang, et al., 2003). The poor bonding linkage between the fillers and the polymer matrix such as the composites made by simple mixing (Hsu, et al., 2003; Liu, et al., 1999; Park, et al., 2003; Sorensen, 2001; Zhang, et al., 1998) will introduce artificial defects, which consequently result in an adverse effect on the mechanical properties of the nanocomposites. Introducing good linkages between the fillers and the polymer matrix is still a challenge for specific composite fabrication. However, appropriate chemical engineering treatment of the nanofiller surface by introducing proper functional groups could improve both the strength and toughness of the subsequent composites with improved compatibility between the nanofillers and the polymer matrix, and make the nanocomposites stable in harsh environments as well. Thus, surface functionalization of nanoparticles with a surfactant or a coupling agent is important not only to stabilize the nanoparticle during processing but also to render them compatible with the polymer matrix. In this paper, the effect of particle functionalization by a functional methacryloxypropyl on the polyester resin curing process and the optimum conditions for high-quality nanocomposite fabrication were investigated. The f-Fe₂O₃ nanocomposites reinforced polyester nanocomposites showed enhanced mechanical properties under tensile, impact and damping study. The nanocomposites containing the f-Fe₂O₃ nanocomposites showed improved thermal stability as compared with the nanocomposites reinforced with the as-received Fe₂O₃ nanocomposites. The nanoparticles were found to be magnetically harder (larger coercivity) after the nanoparticles were dispersed in the polyester resin matrix.

In the present paper, Fe₂O₃, f-Fe₂O₃ nanoparticles were synthesised by chemical reduction method. The major objective of this paper is to explore a novel approach on mechanical properties of two different nanoparticles viz. Fe₂O₃, f-Fe₂O₃ dispersed in to the polyester nanocomposites, i.e. polyester/Fe₂O₃, polyester/f-Fe₂O₃ nanocomposites were developed and characterized. Thus, improved mechanical properties would be expected.

2. MATERIALS

Unsaturated polyester (UP) with 2% cobalt accelerator, catalyst 50% methyl-ethyl-ketone-peroxide (MEKP) and 10% DMA solution was employed in the present study supplied by Dipol 101 from Divine Petrochemicals Pvt. Ltd., India. Ratio of UP/accelerator/catalyst/promoter is 100/2/2/2 (Hsu, et al., 2003; Liu, et al., 1999). The liquid resin has a density of 1.231 g/cm³ and a viscosity of 370 centipoises (cps) at room temperature. Fe₂O₃ nanoparticles with an average diameter of 10-15 nm and a specific surface area of 45m²/g were functionalized and used as nanofillers for the nanocomposite fabrication. Trigonox 239-A (curing catalyst or initiator, organic peroxide, liquid) was purchased from Akzo Nobel Chemicals. Cobalt naphthenate (CoNap, OM Group, Inc.) was used as a catalyst promoter (accelerator) to decompose the catalyst at room temperature. Methacryloxypropyl-trimethoxysilane (MPS) and tetrahydrofuran (THF, anhydrous) were purchased from Sigma-Aldrich Chemical Company (Sawada et al., 2008). All the chemicals were used as-received without further treatment.

2.1. Characterization

Curing characteristics of polyester and polyester filled Fe₂O₃ & f-Fe₂O₃ nanocomposites was studied using DSC. The sample (consisting of resin, Fe₂O₃ & f-Fe₂O₃ and hardener prior to curing)

of 5 mg is taken in an alumina crucible and heat is applied at the rate of 10° C min. Modal analysis is performed to calculate natural frequency and damping factor 'n'. Damping factor is calculated using impulse hammer technique (IHT) and logarithmic decrement method (LDM). Specimens of 250 mm x 25 mm x 4 mm is prepared, in which one end of beam is clamped and other end is attached to accelerometer to obtain vibration modes. Natural frequency is determined by impulse loading at free end of the beam using impulse excitation (Rion PH 7117, modally tuned hammer). The signal received from accelerometer is displayed in Fast Fourier Transform (FFT), in which natural frequency is noted down for various modes. Damping factor 'n' using IHT is determined using half power bandwidth method. The expression for damping factor by half power width technique is given by ($\xi = \Delta\omega/2\omega_n$), where $\Delta\omega$ is bandwidth at half-power points of resonant peak for the nth mode and ω_n is resonant frequency. The half power points are found at $1/\sqrt{2}$ of maximum peak value. In the LDM, sine wave signal is supplied to drive the modal exciter to excite the cantilever beam specimen (Castro, et al., 2000; Wetzel, et al., 2003). During the natural frequency mode, amplitude increases to a large extent, once the resonance is achieved. At this point, the excitation signal is disconnected freely and a typical free decay curve is obtained. From this decay curve, two experimental amplitude data points are collected namely x_1 and x_{n+1} , and the damping factor n is calculated by using the expression

$$\delta = \frac{1}{n+1} \ln \frac{x_1}{x_{n+1}}$$

$$\xi = \frac{\delta^2}{\sqrt{A\pi^2 + \delta^2}}$$

Where ξ , the damping factor, $n+1$, the number of cycles, δ is the logarithmic decrement, x_1 and x_{n+1} are the two displacement values at the time intervals t_1 and t_2 , respectively. A Jeol JSM 840A Japan scanning electron microscope (SEM) was used to study the morphology of fractured surfaces of after tensile test of Fe_2O_3 & f- Fe_2O_3 nanocomposites with different magnifications. Samples were gold-coated initially subjected to SEM analysis. The scanning electron micrographs of different cross-sections of the nanocomposites of pure polyester filled with Fe_2O_3 & f- Fe_2O_3 nanocomposites were studied. Tensile strength was studied by using universal testing machine (UTM) supplied by Instron Corporation; 3369, series-9 was used with a crosshead speed of 5mm/min. Testing samples were prepared like dumb-bell shapes and its dimensions are (100 x 20 x 3 mm³) prepared based on the ASTM D 638 standards. In each case, five samples were tested and the average value tabulated. The thermal characteristics of the Fe_2O_3 & f- Fe_2O_3 filled with polyester nanocomposites were measured using both differential scanning calorimetry (DSC-2010 TA Instrument) and thermogravimetric analyses (TGA) at a rate of 10°C/min under nitrogen flow.

2.2. Preparation of ferrite-nanoparticles

Reaction was carried out by mixing two solutions, named as solution A (i.e. $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$) and solution B (i.e. $\text{FeCl}_2 \cdot 4\text{H}_2\text{O}$) to form nanoparticles. 0.1 Mole of 27.030gms of solution A was dissolved 1000ml deionised water, whereas, 19.881gms of solution B was dissolved in 1000ml deionised water used were to prepare solutions. Then solution A was added then to solution B drop wise by 2:1 stoichiometric ratio under vigorous stirring. Ammonium (45ml) was dropped in to the mixture; forthwith black powder was produced in less than one second. After reaction, the product particles were separated from the solution by a strong magnet and washed with deionised water. Freeze-drying overnight was utilized to dry the particles.

2.3 Surface functionalization of iron oxide nanoparticles

The nanoparticles functionalization follow procedures similar to our earlier reported nanoparticle process (Guo,et al.,2007a; Guo,et al.,2007b) and is described as follows. Fe_2O_3 nanoparticles (13.38 g, 83.8mol) were added into a mixture of 4 g Methacryloxypropyl and 30 ml THF. The resulting colloidal suspension was ultrasonically stirred (Branson 1510) for 1 h and

precipitated by a permanent magnet at room temperature. The precipitated nanoparticles were rinsed with THF to remove excessive Methacryloxypropyl for subsequent nanocomposite fabrication and the dried nanoparticles were used for further particle characterization. The as-received Fe_2O_3 nanoparticles or MPS functionalized Fe_2O_3 nanoparticles were dispersed into 30 ml resin on a specific weight percentage basis. The dispersion was carried out in an ice-water ultrasonic bath for about 1 h. The above particle-suspended solution was then ultrasonically stirred in an ice-water ultrasonic bath until the temperature was cooled down. Then, 2.0 wt% catalyst (initiator) was added into the iron-nanoparticles which was stirred and degassed for 2 min. Next, 0.3 wt% promoter was added and mixed quickly. The mixed viscous solution was poured into silicone rubber molds. The curing via free-radical bulk copolymerization or homopolymerization initiated by the catalyst was done at 85 °C for 1 h under normal atmospheric conditions and cooled down to room temperature naturally in the oven following procedures similar to those used for our reported alumina nanoparticles filled polyester resin nanocomposites fabrication (Judeinstein and Sanchez, 1996). A viscous liquid solution was still observed in the mold after this curing process, indicating that nanofilling materials have a significant effect on the quality of fabricated nanocomposites. However, high-quality nanocomposites were formed by room temperature curing for 24 h followed by post curing at 100 °C for 2 hours.

3. RESULTS AND DISCUSSIONS

The tensile fracture surfaces of polyester and polyester filled f- Fe_2O_3 are shown in Figure 1. It is learnt from the cross section of fractured surface of pure polyester is smooth due to brittle failure. However, on addition of f- Fe_2O_3 particles, crack surface becomes rough. The roughness increases as f- Fe_2O_3 content increases in the matrix. The fracture roughness indicates that the resistance of propagation of crack is small and the crack has not propagated as easily as seen in pure polyester. The fracture surface roughness indicates that crack propagation is large and increased the torturous path of propagating crack (Kohli, et al., 2005; Judeinstein and Sanchez, 1996). This effect results in higher stress to failure and caused improved strength of f- Fe_2O_3 nanocomposites. Though the fracture roughness is predominant at 6% f- Fe_2O_3 , the existence of unexfoliated aggregates, voids, etc. could have decreased the strength of f- Fe_2O_3 . The fracture surface of polyester with Fe_2O_3 series is seen in Figure 2. Fracture surface of polyester with 1% Fe_2O_3 is rougher than pure polyester. At 6% f- Fe_2O_3 , the presence of voids is noted. This indicates that particles have peeled off from material as crack propagates, and create void at the positions where Fe_2O_3 particles were present. This also indicates that bonding between matrix and Fe_2O_3 particle is poor. From both Figures, it is learnt that at 3wt. % f- Fe_2O_3 the roughness was more predominant in the Fig.1 than the Fig.2. In the Fig.2 at 6 wt.% Fe_2O_3 , though the fracture surface is rough though, due to the existence of voids is clearly visible and has decreased the strength of the material. The poor bonding strength, smooth fracture surface, voids, etc. could decrease the tensile strength of the Fe_2O_3 filled polyester nanocomposites. In polyester/ Fe_2O_3 series, though the tensile strength increases up to 3% Fe_2O_3 , for higher nanoparticles content, it decreases due to unexfoliated aggregates, voids, etc. However, the strength of f- Fe_2O_3 nanocomposites are significantly improved than the Fe_2O_3 nanocomposites. It requires further investigation of the synthetic procedure to understand the methods of improving tensile strength for higher nanocontents. Figure 3 shows SEM picture of polyester with 6% f- Fe_2O_3 consisting of voids.

DSC scans of polyester filled with f- Fe_2O_3 , polyester filled with Fe_2O_3 nanocomposites are shown in Figure 4. At 1 wt% f- Fe_2O_3 addition, the peak position was decreased to lower temperature as shown in Figure 4a. At higher wt% f- Fe_2O_3 nanodispersion two distinct peaks were noticed, one at lower temperature and another at higher temperature. The higher temperature peak is noticed at curing peak of 1wt% f- Fe_2O_3 nanocomposites. The lower temperature peak corresponds to curing of polyester resin in intergallery region, and higher temperature peak corresponds to curing at extragallery (matrix) region. As f- Fe_2O_3 content increases above 3 wt% the lower temperature peak decreases and for 6 wt% f- Fe_2O_3 it is at 420°C. Single exothermic peaks was noted for polyester with f- Fe_2O_3 content up to 3 wt% f- Fe_2O_3 and results in uniform polymerization.

This uniform polymerization at intergallery region and extragallery (matrix) region could have favoured the exfoliated structure. At higher Fe_2O_3 content (i.e. 3 wt%), the uneven polymerization occurs in the nanocomposites and this leads to two distinct curing exothermic peaks. The intergallery polymerization occurs at lower temperature and cures quicker than that of the extragallery polymerization. Due to this quick intergallery polymerization, the iron nanolayers cannot separate further. This uneven polymerization leads to the formation of intercalated structures for higher f- Fe_2O_3 content (above 3 wt%). The DSC curing scans of polyester filled with Fe_2O_3 are shown in Fig. 4b. It is seen that the addition of Fe_2O_3 does not shift the exothermic peak, which is noted for f- Fe_2O_3 contents in the Fig.4a. The result suggests that addition of Fe_2O_3 in polyester resin does not affect the curing of polyester. The addition of f- Fe_2O_3 and Fe_2O_3 fillers in the polyester decreases the intensity of exothermic peak. The addition of Fe_2O_3 continuously decreases the exothermic peak and this is due to decrease in concentration of polyester resin on the addition of nanoparticles. It is stated that the presence of organ ions in f- Fe_2O_3 increases the polymerization of polyester by catalytic effect attributed the increase of curing temperature (Rancourt and Taylor, 2008). However, the existence of f- Fe_2O_3 affects the polymerization of polyester. Results shows that the effect of f- Fe_2O_3 restricting the polymerization is more than the polymerization of Fe_2O_3 with polyester polymer, and hence decreases the curing temperature of polyester resin.

Figure 5 shows TGA curves of polyester filled with two different nanoparticles such as Fe_2O_3 and f- Fe_2O_3 nanocomposites. Figure 5 (a) shows thermal stability of 1wt.% and 3wt.% Fe_2O_3 nanocomposites does not show much difference between the both wt.% ratios as both the curves merged up to 300°C and it is due to dissociation of iron oxide ions. It is also learnt that due to the moisture content in the Fe_2O_3 nanoparticles the decomposition was started at 100°C. On addition of 6wt% Fe_2O_3 in polyester, the thermal stability was increased up to 325°C. The hard iron-oxide layers act as barrier for volatile degradation of polyester matrix. This effect causes such enhanced thermal properties of nanocomposites. The enhanced thermal stability of Fe_2O_3 nanocomposites was observed even at higher temperature (>400 °C) was due to the existence of inorganic phases (SiO_2 , Al_2O_3 , MgO , etc) (Park, et al., 2003; Sorensen, 2001) dominates the iron-nanocomposites. Figure 5b shows decomposition of polyester filled with f- Fe_2O_3 nanoparticles. From the Fig.5b shows weight loss was remain constant up to 150°C. This is the clear indication, through functionalization of iron-oxide nanoparticles moisture content is reduced some extent. And also thermal stability was increased to 305°C for 1wt.% f- Fe_2O_3 whereas for 3 and 6wt.% f- Fe_2O_3 up to 330°C and 335°C respectively. Another significant point come to light is decomposition temperature for f- Fe_2O_3 nanoparticles were changed dramatically than Fe_2O_3 nanoparticles. There is a marginal shift in decomposition temperature when f- Fe_2O_3 is added in to the polyester. The improved thermal stability is noticed for polyester filled f- Fe_2O_3 series is higher temperatures (i.e.335°C) than pure iron-oxide nanoparticles.

Figure 6 (a) and (b) shows the effect of addition of Fe_2O_3 and f- Fe_2O_3 nanoparticles filled polyester nanocomposites on tensile strength and tensile modulus. The tensile strength of pure polyester is 41.1 MPa which can be seen in the Fig.6 (a). It is increased up to 3 wt% for both composites, decreases when further dispersion. On other hand, tensile strength was increased up to 3wt.%, decreases in further addition of f- Fe_2O_3 filled polyester nanocomposites. Tensile strength was optimized at 3wt.% for Fe_2O_3 and f- Fe_2O_3 . When nanoparticles are functionalized, it left off exfoliated structures at 3% is the clear indication of higher performance for the f- Fe_2O_3 when compared with Fe_2O_3 . The formation of intercalated structure, agglomeration, presence of voids, etc., above 3% iron-oxide has decreased the strength of the nanocomposites. In the Fig. 6(b), tensile modulus is increased up to 3.5wt.% ,decreases further increase. On the other hand, the tensile modulus of Fe_2O_3 and f- Fe_2O_3 nanocomposites increased with particle content, no matter the specimens were functionalized or not, as shown in Fig. 6(b). This is because the surface modified or not modified nanoparticles reinforced the composites successfully (Zhang, et al., 1998). The Fe_2O_3 particles did not make the remarkable difference in tensile modulus at a lower given particle content. This is because the tensile modulus is measured at small deformation. At this stage, the Fe_2O_3 molecules do not respond to the deformation. But at higher filler content, due to the increase in

viscosity of the particle causes agglomerations, and void induces poor stress transfer, therefore, beyond 3.5wt.% tensile modulus was decreased. The orientation of iron platelets and polymer chains with respect to loading direction can also contribute to reinforcement effects. The decreasing rate of modulus at higher Fe_2O_3 content ($>3.5\%$ Fe_2O_3) is due to presence of not exfoliated aggregates in polyester matrix. In polyester/ f- Fe_2O_3 composites, there was not much improvement in modulus due to the absence of homogeneous Fe_2O_3 distribution at molecular level, and hence does not contribute to molecular strength. The tensile fracture surfaces of polyester filled f- Fe_2O_3 and Fe_2O_3 are shown in Fig. 1. Fig. 2 shows the fracture surface of polyester filled Fe_2O_3 and f- Fe_2O_3 particles. If it seen from Fig. 1 that fracture surface of pure polyester is smooth due to brittle failure. However, on addition of Fe_2O_3 particles, crack surface becomes rough (i.e. ductile nature). The roughness increases as Fe_2O_3 content increases in the matrix. The fracture roughness indicates that the resistance of propagation of crack is large and the crack has not propagated as easily as seen in pure polyester. The fracture surface roughness indicates that crack propagation is large and increased the torturous path of propagating crack (Chiang, et al., 2003). This effect results in higher stress to failure and caused improved strength of Fe_2O_3 -nanocomposites.

3.1 Impact properties

Impact results of polyester filled with Fe_2O_3 and f- Fe_2O_3 nanocomposites are shown in Fig. 7. It is seen that impact strength is increased up to 2.5wt.% Fe_2O_3 and further addition of Fe_2O_3 decreases the impact strength. The decrease in impact strength at higher filler content is due to the existence of agglomeration, not exfoliated aggregates, voids, etc. The addition of f- Fe_2O_3 in polyester decreases the impact strength of polyester/nanocomposites beyond 3%. The impact strength results shows that f- Fe_2O_3 nanocomposites provide better impact properties than Fe_2O_3 filled polyester composites. Figure 1, 2, and 3 shows the impact fracture surface of nanocomposites. The impact fracture surface provides the reason for impact properties in nanocomposites. Figure 1 shows that fracture surface of pure polyester is smooth and was indicating brittle failure. The addition of 1% Fe_2O_3 , fracture surface shows rough morphology. The roughness of the fracture surface increases as Fe_2O_3 content increases to 2.5% Fe_2O_3 . The existence of rough surface shows that crack propagation is difficult and could have increased the torturous path and leads to high strength to failure. This has caused high impact strength of Fe_2O_3 nanocomposites up to the addition of 2.5 wt% Fe_2O_3 , and on higher addition. Impact results suggest that some additional energy absorbing mechanism is taking place when f- Fe_2O_3 particles are reinforced in matrix. Crack pinning, cavitations mechanisms, crack surface roughness, etc. (Gao and Mader, 2002) are the possible reasons for high impact strength for f- Fe_2O_3 filled polyester higher than that of Fe_2O_3 filled polyester, hence **Table 1** shows the effect of Fe_2O_3 addition on natural frequency of polyester. Pure polyester shows natural frequency of 17.56, 118.53, 120.53 and 298.34Hz for 1st four modes of vibrations, respectively. On addition of Fe_2O_3 , natural frequencies of all modes are higher than the polyester. For higher Fe_2O_3 content (>3 wt %), natural frequency decreases, however higher than that of pure polyester. The large increase in stiffness due to the reinforcement of Fe_2O_3 in the matrix causes such increased natural frequency. For higher Fe_2O_3 content, the presence of not exfoliated aggregates of Fe_2O_3 reduces the stiffness, which decreases the natural frequency. Table 2 shows the effect of f- Fe_2O_3 addition on natural frequency of polyester. The Fe_2O_3 addition does not improve the natural frequency of polyester as is seen in the case of f- Fe_2O_3 filler additions. A negligible effect in stiffness on reinforcement of Fe_2O_3 in polyester polymer matrix causes such low natural frequencies. Figure 8 (a) and (b) shows the effect of Fe_2O_3 and f- Fe_2O_3 addition on damping characteristics of polyester nanocomposites. Damping factors measured by LDM and IHT methods for 1st and 4th mode of natural frequencies are presented. Damping factors measured for 1st mode of natural frequencies of epoxy with Fe_2O_3 and f- Fe_2O_3 series are shown in Fig. 8a. It is observed that both the Fe_2O_3 and f- Fe_2O_3 addition increases the damping factors of pure polyester. The Fe_2O_3 filled polyester increases the damping factor than that of f- Fe_2O_3 filled polyester polymer. Damping factor measured by IHT shows higher values than that of measured by LDM. The free load during impact causes increased damping in IHT. Since no free load is acting in LDM, and hence damping

factor is less than that of IHT. It is seen that damping factor increases up to 3 wt% of Fe_2O_3 , and for higher Fe_2O_3 addition in polyester, damping factor decreases but above the value of matrix material. The increased stiffness due to the addition of Fe_2O_3 improves damping factor (Liu, et al., 1999). Similar effect in damping is noted for 4th mode of natural frequency of epoxy filled Fe_2O_3 and f- Fe_2O_3 series as shown in Fig. 8(b). Though the addition of f- Fe_2O_3 shows improvement in damping factor, the values are lesser than that of Fe_2O_3 filled polyester polymer.

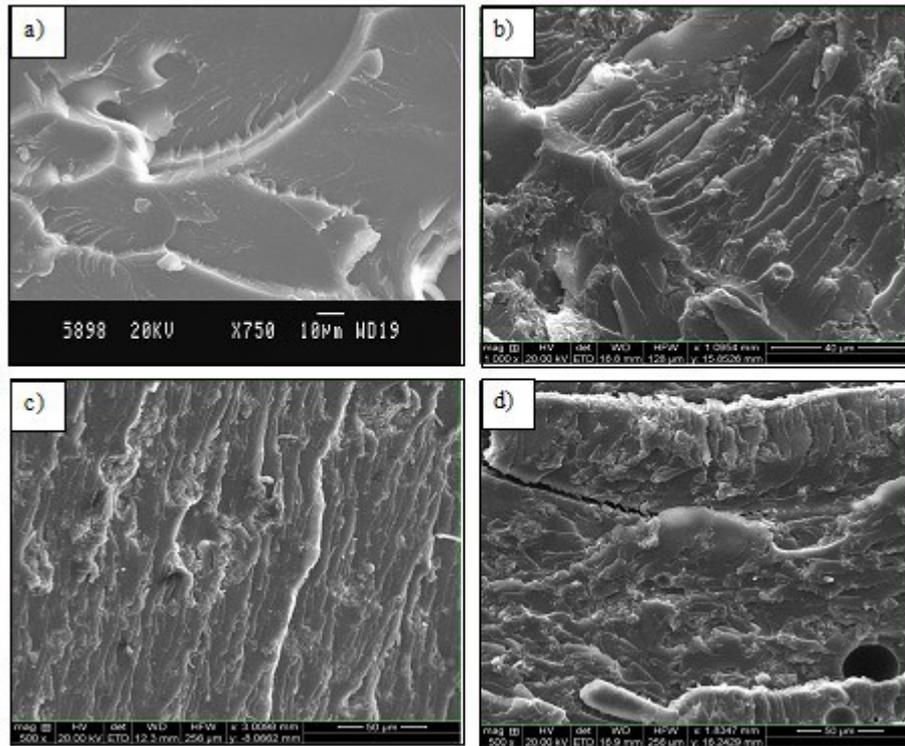


Figure 1 SEM analysis on tensile fracture surface of (a) pure polyester (P) (b) P + 1% f- Fe_2O_3 , (c) P + 3% f- Fe_2O_3 and (d) P+ 6% f- Fe_2O_3 nanocomposites.

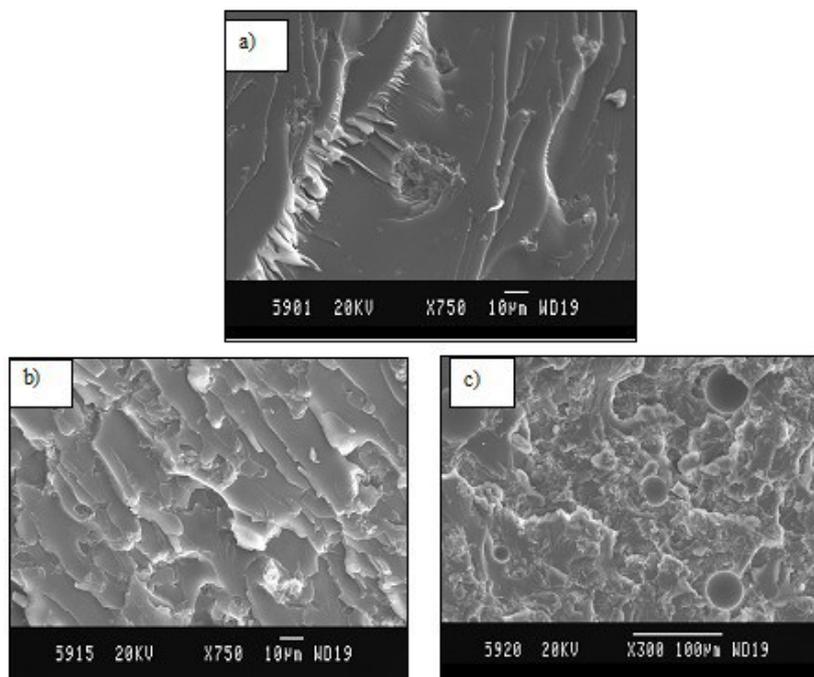


Figure 2 SEM analysis on tensile fracture surface of (a) Polyester (P) + 1% Fe_2O_3 , (b) P + 3% Fe_2O_3 and (c) P + 6% Fe_2O_3 .

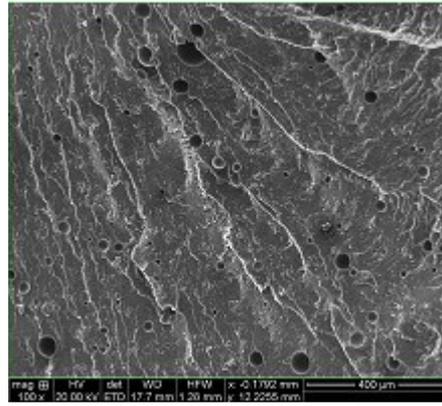


Figure 3 SEM analysis on tensile fracture surface of P + 6% f-Fe₂O₃ nanocomposites.

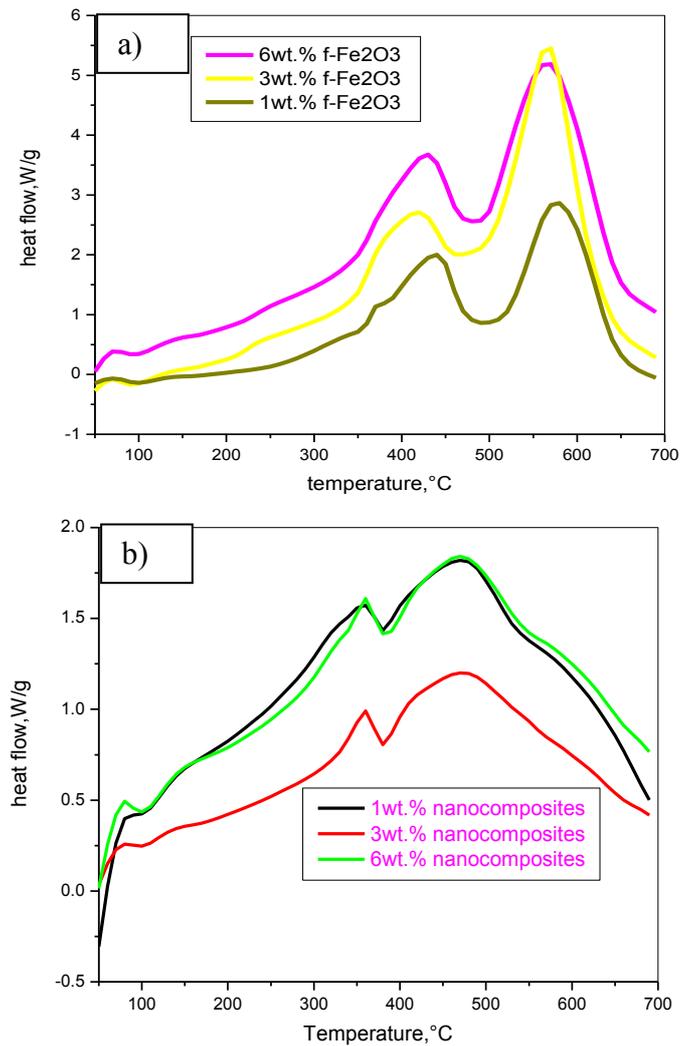


Figure 4. DSC scans of (a) polyester with f-Fe₂O₃ series (b) Polyester with Fe₂O₃ series.

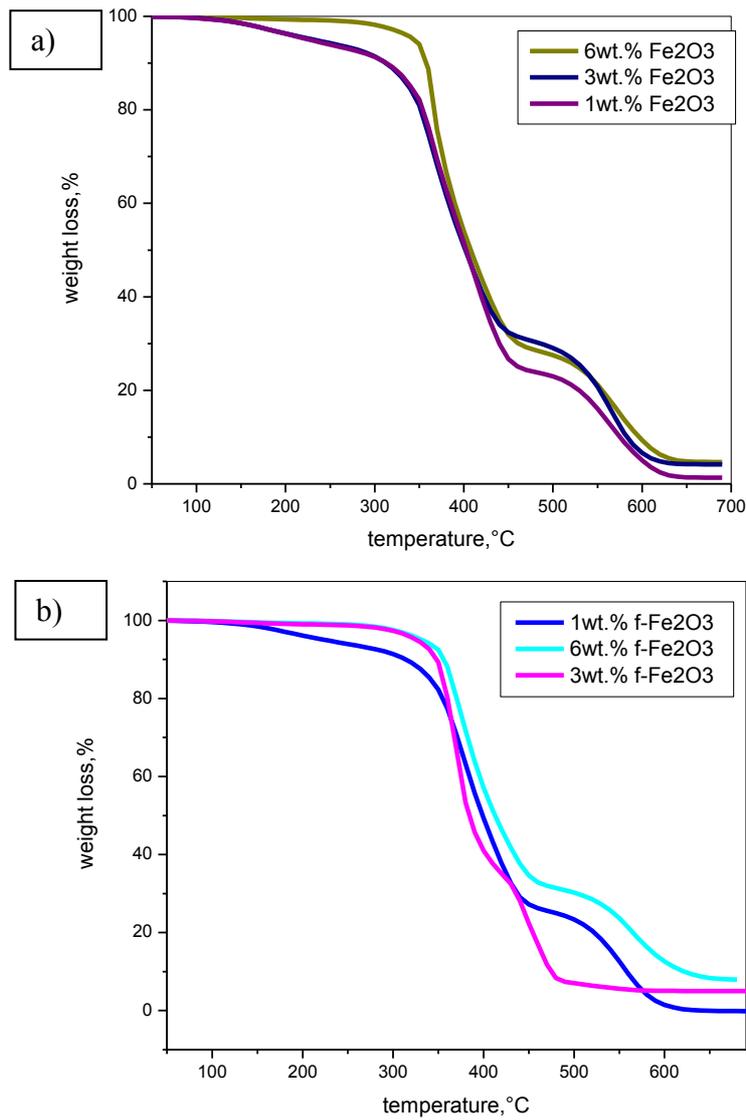
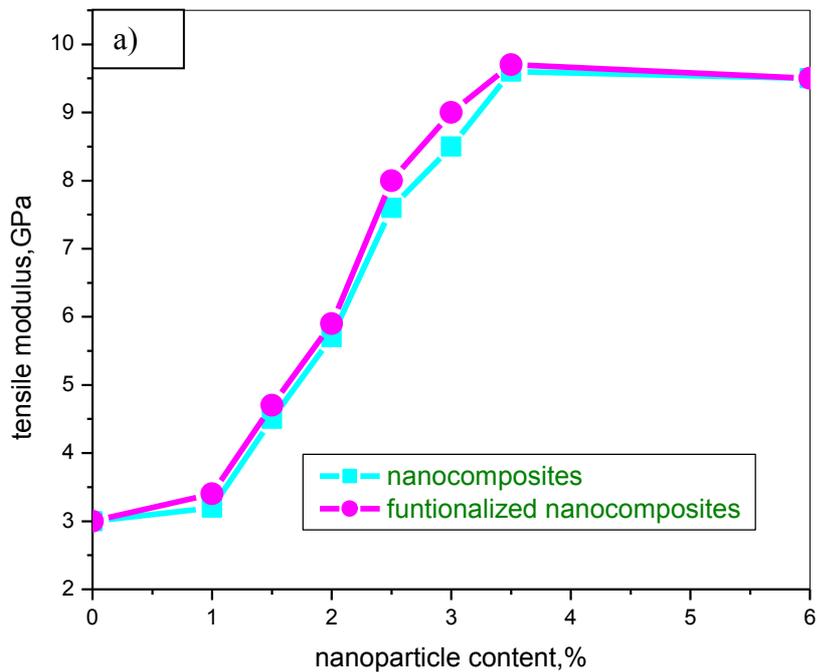


Figure 5. TGA thermographs of polyester with (a) Fe₂O₃ and (b) f-Fe₂O₃ composite series.



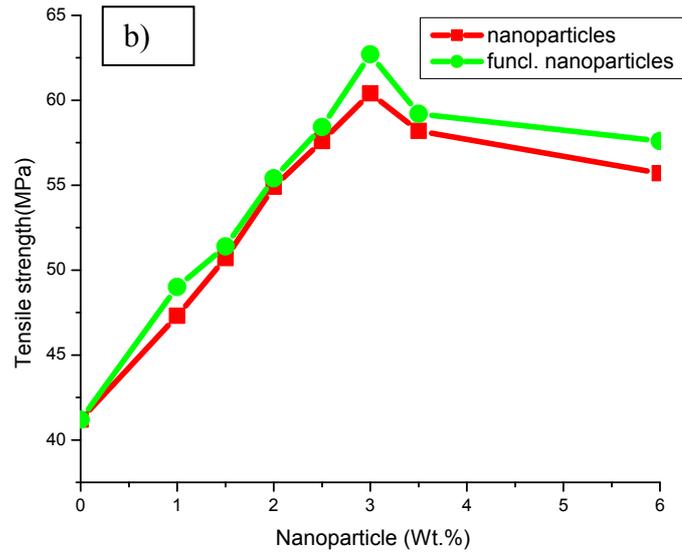


Figure 6 Effect of Fe_2O_3 / $\text{f-Fe}_2\text{O}_3$ content on (a) tensile strength and (b) tensile modulus

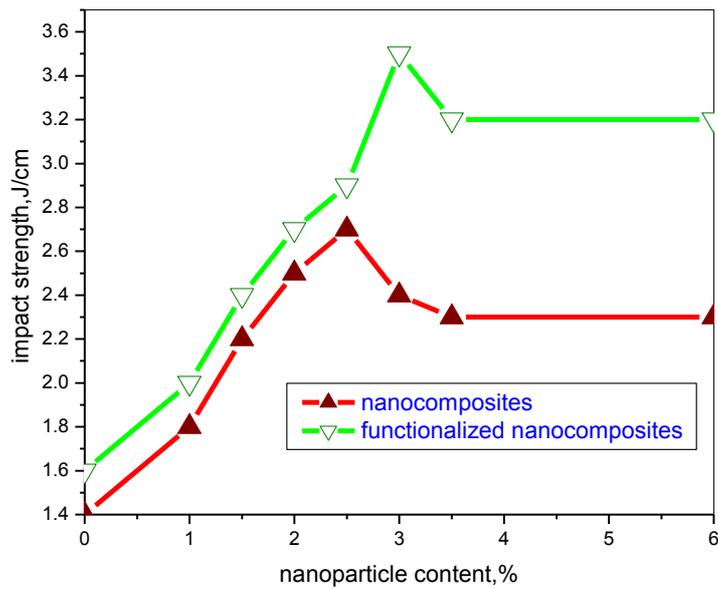


Figure 7 Effect of impact strength of polyester filled with Fe_2O_3 and $\text{f-Fe}_2\text{O}_3$ nanocomposite series.

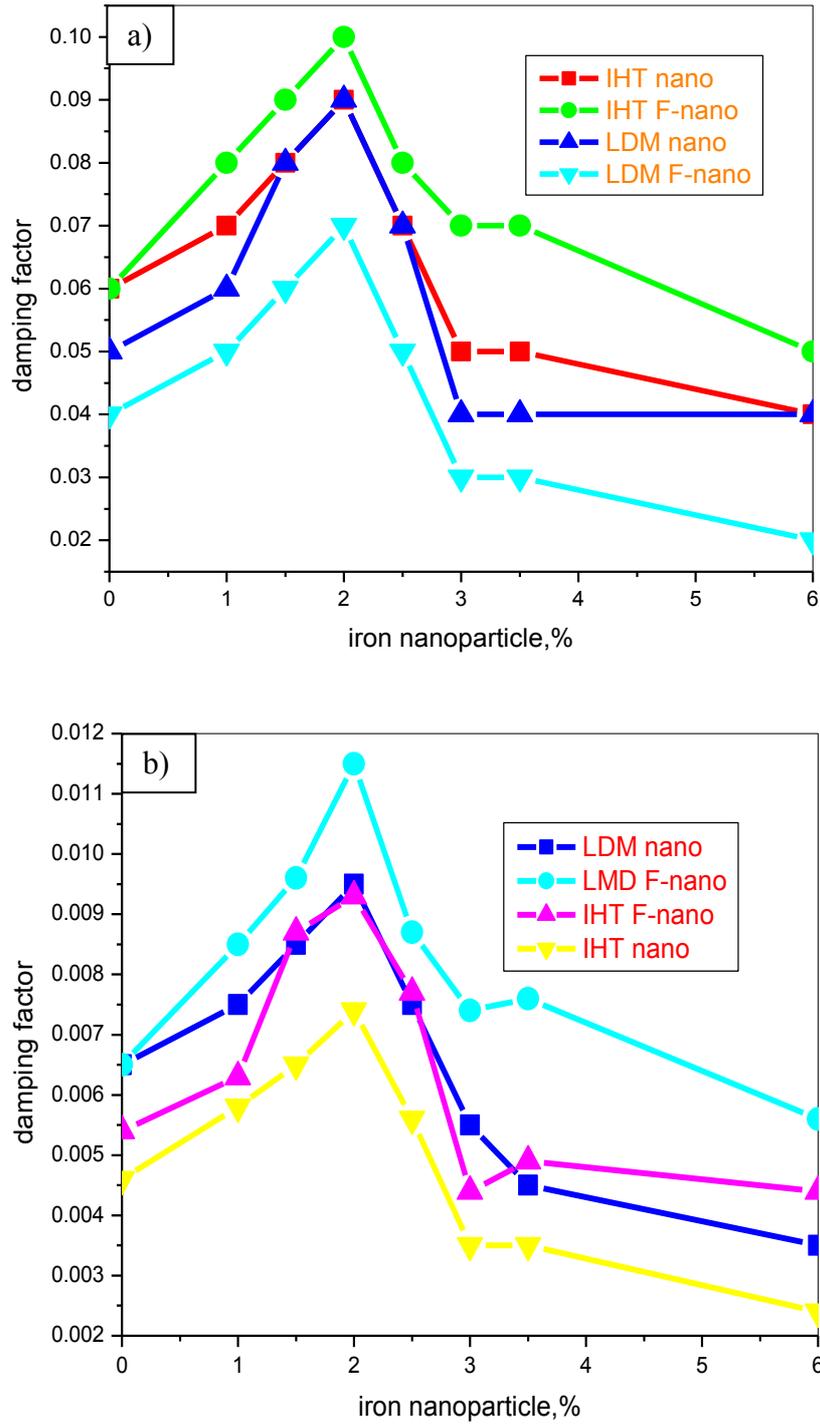


Figure 8 Damping factor for polyester with Fe_2O_3 and $f-Fe_2O_3$ series at (a) 1st mode and (b) 2nd mode.

Table 1 Frequency dependence of polyester filled with nanoparticles (f-Fe₂O₃) at four different modes.

(Polyester + wt.% f-Fe ₂ O ₃)	Natural Frequency at mode 1, Hz	Natural Frequency at mode 2, Hz	Natural Frequency at mode 3, Hz	Natural Frequency at mode 4, Hz
P+0.0	14.66	100.10	140.00	195.00
P+1.0	17.23	122.07	153.75	200.05
P+1.5	19.51	124.96	161.75	230.42
P+2.0	20.40	135.08	184.63	248.43
P+2.5	21.20	137.04	214.06	258.26
P+3.0	23.24	139.50	220.75	278.43
P+3.5	18.23	130.45	201.43	261.23
P+6.0	18.01	131.09	200.55	260.34

Note: P-Polyester

Table 2 Frequency dependence of polyester filled with nanoparticles (Fe₂O₃) at four different modes.

(Polyester + wt.% Fe ₂ O ₃)	Natural Frequency at mode 1, Hz	Natural Frequency at mode 2, Hz	Natural Frequency at mode 3, Hz	Natural Frequency at mode 4, Hz
P+0.0	14.66	100.10	140.00	195.00
P+1.0	15.02	115.60	145.02	196.48
P+1.5	15.99	116.16	147.32	200.43
P+2.0	16.40	120.29	149.42	210.75
P+2.5	17.42	130.48	170.54	224.23
P+3.0	18.71	131.70	190.25	235.40
P+3.5	17.01	129.46	185.35	230.09
P+6.0	16.52	127.88	182.04	225.77

Note: P-Polyester

4. CONCLUSIONS

Two highly nanodisperse Fe₂O₃ & f-Fe₂O₃ nanoparticles were synthesized through chemical reduction method and then dispersed into the polyester polymer. Tensile strength, tensile modulus and impact strength and damping characteristics were studied for Fe₂O₃ & f-Fe₂O₃/polyester nanocomposites. From the SEM analysis it is learnt that addition of Fe₂O₃ & f-Fe₂O₃ nanoparticles impart the changes in the phase transformation from brittle to ductile nature that ultimately rule the improved performance up to optimum weight %, on other hand, beyond optimum level performance is reduced due to the formation of agglomerations and voids due to the increased viscosity of the modified polymer solution. Mechanical properties like tensile strength, modulus and impact strength were optimized due to the same reason as mentioned above. In DSC and TGA analysis it is learnt that addition inorganic nanosuspension in to the polymer were played their role in improvement of thermal stability and curing characteristics. Nanoparticles of iron oxides nanocomposites can be used as ferrofluids, high-density information storage, magnetic resonance imaging (MRI), tissue-specific releasing of therapeutic agents, biomechanical applications.

Acknowledgements

Authors would like to thank the Department of Polymer Science and Technology, S.K. University, Anantapur and RV College of Engineering, Bangalore, India for giving their support in completion of our research.

References

- [1] Chen, Y.; Sun, L.; Chiparus, O.; Negulescu, I.; Yachmenev, V.; Warnock, M.(2005). Kenaf/ramie composite for automotive headliner. *J Polym. Envint.***2005**, 13,107–114.
- [2] Podlaha, E.J.; Landolt, D. Pulse-reverse plating of nanocomposite thin films. *Journal of Electrochemical Society*, **1997**, 144, 200–202.
- [3] Castro, C.; Ramos, J.; Millan, A.; Gonzalez-Calbet, J.; Palacio, F. Production of magnetic nanoparticles in imine polymer matrixes. *Chemical Materials*, **2000**, 12, 3681–3688.
- [4] Wetzel, B.; Hauptert, F.; Zhang, M.Q. Epoxy nanocomposites with high mechanical and tribological performance. *Composite Science and Technology*, **2003**, 63, 2055–2067.
- [5] Guo, Z.; Park, S.; Wei, S.; Pereira, T.; Moldovan, M.; Karki, A.B. Flexible high-loading particle-reinforced polyurethane magnetic nanocomposite fabrication through particle-surface-initiated polymerization. *Nanotechnology*, **2007**, 18,335-340.
- [6] Kohli, N.; Worden, R.M.; Lee, I. Intact transfer of layered, bionanocomposite arrays by microcontact printing. *Chemical Communications*, **2005**, 3, 316–318.
- [7] Judeinstein, P.; Sanchez, C. Hybrid organic–inorganic materials: a land of multi-disciplinarity. *Journal of Materials Chemistry*, **1996**, 26, 511–525.
- [8] Zhang, X.; Simon, L.C. In situ polymerization of hybrid polyethylene– alumina nanocomposites. *Macromolecular Material Engineering*, **2005**, 290,573–583.
- [9] Gao, S.L.; Mader, E. Characterization of inter phase nanoscale property variations in glass fibre reinforced polypropylene and epoxy resin composites. *Composites A*, **2002**, 33:559–576.
- [10] Shenhar, R.; Norsten, T.B.; Rotello, V.M. Polymer-mediated nanoparticle assembly: structural control and applications. *Advanced Materials*, **2005**, 17, 657–669.
- [11] Guo, Z.; Pereira, T.; Choi, O.; Wang, Y.; Hahn, H.T. Surface functionalized alumina nanoparticle filled polymeric nanocomposites with enhanced mechanical properties. *Journal of Materials Chemistry*, **2006**, 16, 2800–2808.
- [12] Guo, Z.; Wei, S.; Shedd, B.; Scaffaro, R.; Pereira, T.; Hahn H.T. Particle surface engineering effect on the mechanical, optical and photoluminescent properties of ZnO/vinyl-ester resin nanocomposites. *Journal of Material Chemistry*, **2007**,17, 806–813.
- [13] Guo, Z.; Liang, X.; Pereira, T.; Scaffaro, R.; Hahn, H.T. CuO nanoparticles filled vinyl-ester resin nanocomposites: fabrication, characterization and property analysis. *Composite Science and Technology*, **2007**, 67, 2036–2044.
- [14] Rancourt, J.D.; Taylor, L.T. Preparation and properties of surface conductive polyimide films via in situ co-deposition of metal salts. *Macromolecules*, **2008**, 20, 790–795.
- [15] Sawada, T.; Ando, S. Synthesis, Characterization, and Optical Properties of Metal-Containing Fluorinated Polyimide Films. *Chemistry Materials*, **2009**, 10, 3368–3378.
- [16] Chiang, P.C.; Whang, W.T. The synthesis and morphology characteristic study of bao-odpa polyimide/tio₂ nanohybrid films. *Polymer*, **2003**, 44, 2249–2254.
- [17] Hsu, S.C.; Whang, W.T.; Hung, C.H.; Chiang, P.C.; Hsiao, Y.N. Effect of the Polyimide Structure and ZnO Concentration on the Morphology and Characteristics of Polyimide/ZnO Nanohybrid Films. *Macromolecular Chemistry Physics*, **2005**, 206, 291–298.
- [18] Liu, L.; Zhang, T.J.; Cui, K.; Dong, Y.D.(1999). Reduction of copper oxide with graphite by mechanical alloying. *Journal of Material Research*, **1999**, 14, 4062–4069.

-
- [19] Park, S.S.; Bernet, N.; Roche, S.D.L.; Hahn, H.T. Processing of iron oxide-epoxy vinyl-ester nanocomposites. *Journal of Composite Materials*, **2003**, 37,465–476.
- [20] Sorensen, C.M. Magnetic Nanoscale Materials in Chemistry. In: Klabunde KJ, editor. *Nanoscale Materials in Chemistry*. New York: Wiley-Interscience; **2001**, 69–221.
- [21] Zhang, D.; Klabunde, K.J.; Sorensen, C.M.; Hadjipanayis, G.C. Magnetization temperature dependence in iron nanoparticles. *Physical Review B*, **1998**, 58,14167–14170.